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# Structure-sensitive scaling relations among carbon-containing species and their possible impact on CO<sub>2</sub> electroreduction

Manuel J. Kolb,<sup>1</sup> David Loffreda,<sup>2</sup> Philippe Sautet,<sup>3,\*</sup> and Federico Calle-Vallejo<sup>1,\*</sup>

<sup>1</sup> Departament de Ciència de Materials i Química Física, Institut de Química Teòrica i Computacional (IQTCUB), Universitat de Barcelona, Martí i Franquès 1, 08028 Barcelona, Spain.

<sup>2</sup> Univ Lyon, Ens de Lyon, CNRS UMR 5182, Université Claude Bernard Lyon 1, F-69342 Lyon, France.

<sup>3</sup> Department of Chemical and Biomolecular Engineering and Department of Chemistry and Biochemistry, University of California, Los Angeles, Los Angeles, California 90095, United States.

\* Correspondence to: <u>sautet@ucla.edu</u>, <u>f.calle.vallejo@ub.edu</u>

### Abstract

The arduous modelling of reactions at heterogeneous catalysts is greatly simplified when adsorption-energy scaling relations between intermediates exist. The offset of these linear relations is structure-independent when the slope is unity and otherwise depends on the coordination number of the active sites. Here we examine the adsorption of \*C, \*CH, \*CH<sub>2</sub>, \*CH<sub>3</sub> and \*COH on five different surface sites of nine transition metals to establish their structure-sensitive scaling relations. Interestingly, we show that the scaling relations of \*C (valency 4) and C-containing species with valency 3 (\*CH, \*COH) have peculiar structure-independent offsets. These offsets stem from the analogous bonding of those adsorbates to the adsorption sites, in spite of their dissimilar valency. We show how this result implies that reaction pathways in catalysis involving \*C, \*CH and \*COH, for instance  $CO_2$  electroreduction to CH<sub>4</sub>, will usually have sizable thermodynamic limits imposed to their optimization.

# Highlights

- DFT-based adsorption energies for C, CH<sub>x</sub> and COH on various transition metal sites
- The slopes of the scaling lines can be predicted based on electron-counting rules
- Anomalous structure dependence for the offsets of CH vs C and COH vs C
- COH hydrogenation to produce C is energy-intensive during CO<sub>2</sub> reduction to CH<sub>4</sub>





#### **1. Introduction**

The release of  $CO_2$  from anthropogenic sources has been increasing by about 1% per year in the last few years [1], despite the current efforts towards its reduction, which has led to a great imbalance in the biogeochemical cycle of carbon. While the forced lockdown during the COVID-19 pandemic has led to a temporary reduction in the emissions by early April of an estimated 17% and a yearly estimated reduction of 4-7% for 2020 [2], long-term solutions are necessary to mitigate climate change. Among the most centralized emitters of  $CO_2$  are the cement industry with around 5-7% of the global emissions [3] and the fossil-fuel-based electric power production, where the 1000 largest plants produce up to 22% of the total global fossil fuel  $CO_2$  emissions [4]. For these centralized, high-volume emitters,  $CO_2$  capture and conversion in electrolyzers is a realistic alternative [5–8]. Furthermore, being able to use renewable energies directly to electrochemically reduce carbon dioxide selectively to valuable fuels and commodities would allow for an efficient carbon recycling scheme [9].  $CO_2$  electroreduction is extremely sensitive to the employed catalyst [10–12] as well as the solvent and the electrolyte used [13,14], which makes the discovery and optimization of catalysts a formidable task [15,16].

An approach to help streamline the search for catalysts is the usage of screening routines based on adsorption-energy scaling relations [17–22]. These relations connect the adsorption energies of two adsorbates A and B on facet i as follow [23,24]:

$$\Delta E_A^i = m \cdot \Delta E_B^i + b^i \tag{1}$$

where the slope *m* is nearly constant for all facets and the y-axis intercept  $b^i$ , hereon referred to as the offset, depends on the geometric structure of the adsorption sites [23,24].

The slope of the scaling line is typically positive, dimensionless, relatively independent of the simulation method employed and can be assessed by simple electron counting rules [23,25]:  $m = N_A/N_B$ , where  $N_A$  and  $N_B$  are the number of valence electrons that A and B miss to fulfill the octet rule. For example, the \*CH vs \*C (we follow here the convention y vs x) scaling relation is predicted to have an slope of 3/4, as \*C lacks 4 valence electrons to reach the octet and the carbon atom in \*CH lacks 3. Likewise, the scaling relation of \*CH<sub>2</sub> vs \*C should have an inclination of 1/2, and that of \*CH<sub>3</sub> vs \*C should be 1/4. While this is generally true [23], it is worth noting that phenomena such as the increasing covalent interaction between surface sites and adsorbates can induce departures from the predicted values [26], and usually reduce the

slope. Besides, negative slopes have also been observed when low-electronegativity species such as \*B (adsorbed boron) are scaled with high-electronegativity species such as \*N [27].

The offset has units of energy and is related to the absolute adsorption strength. Thus, it needs to be calculated using e.g. first-principles methods and explicitly depends on the functional and basis sets used. Scaling relations exist on different materials [28,29] and sites including terraces, steps, kinks and akin undercoordinated surface sites [24,30]. When comparing the offsets across different facets for adsorbed oxygen and oxygenates (\*O, \*OH, \*OOH and \*OCH<sub>3</sub>), it was observed that they are linearly correlated with the coordination number of the sites, as long as the slope is not 1 [24]. This means that the absolute adsorption energies for low-coordinated sites can, in principle, be predicted using comparatively simple calculations on high-symmetry facets such as the (111) if the coordination number of the active sites is known. On the other hand, when the slope is unity, the offset is surface independent and can be easily approximated from gas-phase calculations and/or experiments [24,31].

Adsorption-energy scaling relations among  $*C_xH_y$  species are well known [23,28,32]. Besides, the scaling relation between \*CO and \*CHO has received considerable attention in the literature [33–36] and its breakage is considered essential [16,17] to improve the catalytic processes in which it takes place. However, the structural sensitivity of scaling relations among carbon-based adsorbates remains largely unexplored. Knowledge of that is necessary for the development of swift yet accurate screening routines for  $CO_2$  reduction and several other reactions. In this work, we provide the scaling relations of various carbon-containing adsorbates (\*C, \*CH,  $*CH_2$ ,  $*CH_3$ , \*COH) on nine transition metals and five adsorption sites with different coordination numbers. We found a peculiar behavior of the offset of the \*CH vs \*C and \*COHvs \*C scaling relations, which would only be expected for pairs of adsorbates with unity slopes. Finally, we discuss the reaches of these results in the light of electrocatalytic  $CO_2$  reduction.

### 2. Computational details

The DFT calculations were performed with the VASP code [37], using the projector augmentedwave method to describe ion-electron interactions [38], and the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional [39]. Geometry optimizations were carried out with a planewave cutoff of 450 eV and the conjugate gradient method. We studied the surfaces of Cu, Ag, Au, Ni, Pd, Pt, Co, Rh and Ir. All metals were simulated in their fcc phase. The tests in section S3 show the reliability of this approximation for Co. Previous works showed that although including spin polarization affects the actual adsorption energies for Co and Ni, the changes in the parameters of the scaling relations are negligible [35], which is corroborated in section S3. Thus, as a first approximation, we only provide here results for spin-restricted calculations on Co and Ni slabs.

In view of their unsaturation, \*C, \*CH and \*COH usually bind to three or four surface atoms. Therefore, we selected for this study surfaces containing threefold or fourfold hollow sites. In brief, we used  $(2\times 2)$  (111) slabs with 4 layers,  $(3\times 2)$  (100) slabs with 4 layers,  $(2\times 2)$ (110) slabs with 5 layers,  $(4\times3)$  (100) slabs with 4 layers and 4 metal adatoms (denoted 4AD@100), (3×3) (111) slabs with 4 layers and 3 metal adatoms (denoted 3AD@111), see schematics in Figure 1. All metal atoms in the two topmost layers of the slabs, the metal adatoms, and the adsorbates were allowed to move in all directions until the maximal force on any atom was below 0.05 eV Å<sup>-1</sup>. The tests done in section S2 show the convergence of the calculations in regard to this cutoff criterion and indicate that the convergence criteria used here are adequate. The atoms in the bottommost layers were set at the optimized bulk distances and made immobile during the simulations. In all cases, the unit cells were large enough in the surface plane to avoid lateral interactions between adsorbates, see Figure 1 and Figure S1 in the Supporting Information (SI). At least 12 Å of vacuum were added in the vertical direction between periodically repeated images and dipole moment corrections were considered during the geometry optimizations to correct the total electronic energies from spurious electrostatic interactions between equivalent periodic slabs. The Methfessel-Paxton method [40] was used to smear the Fermi level with  $k_BT = 0.2 \text{ eV}$ , and all energies were extrapolated to 0 K. The k-point samplings used were  $6 \times 6 \times 1$  for (111) slabs,  $6 \times 8 \times 1$  for (100) slabs,  $5 \times 4 \times 1$  for (110) slabs,  $3 \times 4 \times 1$ for 4AD@100, and  $(4 \times 4 \times 1)$  for 3AD@111.

The adsorption energies of species *w* were calculated as:  $\Delta E_w = E_{*w} - E_* - E_w$ , with \*w = \*C, \*CH,  $*CH_2$ ,  $*CH_3$ , \*COH; and w = C(g), CH(g),  $CH_2(g)$ ,  $CH_3(g)$ , COH(g). The total energies of the gas-phase species were calculated in cubic boxes of 15 Å × 15 Å × 15 Å, with  $k_BT = 0.001$  eV, and sampling only the  $\Gamma$ -point in spin unrestricted calculations. The resulting magnetic moments of C(g), CH(g), CH<sub>2</sub>(g), CH<sub>3</sub>(g), COH(g) are 4, 1, 2, 1, and 1. The adsorption

energies of \*COH were adapted from previous works [35], except for those on 3AD@111 sites, which were calculated in this work. The free energies were approximated as:  $G \approx E + ZPE - TS + E_{solvation}$ . *E* is the total energy calculated with DFT; *ZPE* is the zero-point energy calculated with DFT using the harmonic oscillator approximation, T = 298.15 K, *S* is the total entropy for gas-phase species and the vibrational entropy for adsorbates; and  $E_{solvation}$  is the stabilization granted by adsorbate-water interactions, taken from previous works [41], see further details in section S6 and specific values in Table S10. The energetics of proton-electron pairs was calculated with the computational hydrogen electrode [42] and the total energy of CO<sub>2</sub> corrected using the method in ref. [43].



**Figure 1.** Structures of the adsorption sites under study. a) Pt(111), b) Pt(100), c) Pt(110), d) 4AD@Pt(100), e) 3AD@Pt(111). The adsorption sites where adsorption was inspected for each adsorbate are marked as follows: fcc threefold hollow site (fcc), hcp threefold hollow site (hcp), fourfold hollow sites (h), atop site (top), bridge site (br), long bridge on the Pt(110) site (br<sub>2</sub>). The adsorption site preferences with respect to the adsorbate and the surface type can be found in Tables S3-S8 in the SI. Figure S1 shows the unit cells used in the calculations.

#### 3. Results and discussion

## **3.1.** Scaling relations for $CH_x$ species (x = 1, 2, 3)

The five adsorption sites investigated together with their representative coordination number can be found in Figure 1. The adsorption site preferences can be found in Tables S3-S8 in the SI for each species, metal and coordination number. We observed that \*C, \*CH and \*COH prefer to adsorb at three- or fourfold hollow sites in the vast majority of cases, whereas \*CH<sub>2</sub> tends to adsorb at bridge sites, but also adsorbs in some cases on hollow sites. \*CH<sub>3</sub> adsorbs on bridge and top sites.

The scaling relations of  $*CH_x$  vs \*C, established using the most stable adsorption sites for every adsorbate, appear in Figure 2. For the adsorption of \*CH vs \*C we observe an average slope of 0.77, which is close to the expected value of  $\frac{3}{4}$  and in agreement with the values in previous publications [23]. For the  $*CH_2$  vs \*C scaling we observe an average slope of 0.50 which, again, matches the prediction of  $\frac{1}{2}$  based on electron-counting rules and compares favorably to previous results [23] . Lastly, the  $*CH_3$  vs \*C scaling shows an average slope of 0.25, which also matches the prediction of  $\frac{1}{4}$  and previous results [23]. In conclusion, as in the case of oxygenates [24], the slope is not structure sensitive and can, as a first approximation, be safely estimated from electron-counting rules.



**Figure 2.** Adsorption-energy scaling relations between the adsorption energies of \*C and \*CH (red), \*CH<sub>2</sub> (orange), and \*CH<sub>3</sub> (gray) on the five surface sites under study, for nine transition metals. The facet and its representative coordination number (*cn*) is provided in each panel. The parameters of the linear fits can be found in Table S8.

#### 3.2. The offsets of the \*CH<sub>x</sub> vs \*C scaling relations

We collected the offsets of all scaling relations displayed in Figure 2 and plotted them as a function of the coordination number of the adsorption sites in Figure 3. We observe that the offsets for \*CH<sub>2</sub> vs \*C and \*CH<sub>3</sub> vs \*C exhibit a linear growth with increasing coordination number, in line with previous observations for oxygenates vs \*O [24]. As shown in section S5, the ratio of the slopes for \*CH<sub>3</sub> vs \*C and \*CH<sub>2</sub> vs \*C in Figure 3 is connected to the adsorption energies of \*CH<sub>3</sub> and \*CH<sub>2</sub> by means of Equation 2, where  $\alpha_{CH_x vs C}$  is the slope in Figure 3, *i* and *j* are two different surface facets (see Figure 1) with coordination numbers  $cn^i$  and  $cn^j$ , and *M* is a given metal.

$$\frac{\alpha_{CH_3 \text{ vs } C}}{\alpha_{CH_2 \text{ vs } C}} \approx \left(\frac{\Delta E_{CH_3}^{i,M} - \Delta E_{CH_3}^{j,M}}{\Delta E_{CH_2}^{i,M} - \Delta E_{CH_2}^{j,M}}\right)$$
(2)

According to Equation 2, the relatively larger slope observed in Figure 3 for the \*CH<sub>3</sub> vs \*C scaling relation with respect to \*CH<sub>2</sub> vs \*C stems from the fact that \*CH<sub>3</sub> adsorption energies generally become more negative than those of \*CH<sub>2</sub> when the coordination number of the active sites is decreased, see further details in section S5.



**Figure 3.** Offsets of the adsorption-energy scaling relations between \*C and \*CH (red),  $*CH_2$  (orange), and  $*CH_3$  (gray) as a function of the coordination number of the adsorption sites. The parameters of the linear fits can be found in the SI, Table S8.

In contrast, the offset for \*CH vs \*C is essentially constant for all studied surface sites at an average value of -0.98 eV, with a mean absolute error (MAE) of 0.07 eV. In turn, this implies

that the adsorption energy difference between \*C and \*CH adsorption is nearly constant for all facets of a given metal. The constant offset can be rationalized as follows: first, on every facet and metal \*C and \*CH adsorb always at the same sites, namely hollow sites, unlike \*CH<sub>2</sub> and \*CH<sub>3</sub> (see Tables S3-S8). Second, the bonding of \*C is presumably similar to that of \*CH because it is difficult for \*C to make four  $\sigma$  bonds to the surface, in view of its hybridization and the spatial orientation of its orbitals [23]. Indeed, a \*C atom with four bonds has sp<sup>3</sup> hybridization and, because of that, should be spatially organized in a tetrahedral fashion. This is not possible on a flat, 2D adsorption site, where all surface atoms are in the same plane. We note as well that some surface sites in Figure 1 are threefold, such that, following effective medium theory [23,44] \*C can only position itself at a distance from the surface that offers an electron density equivalent to that a stable gas-phase counterpart, namely, CH4.

Accordingly, the constant energetic separation could be a general phenomenon noticeable in the scaling relations between \*C and C-containing species with a valency of 3. To evaluate this hypothesis, we calculated the adsorption energies of \*COH, the carbon atom of which lacks three electrons to fulfill the octet rule, similar to \*CH (see Tables S3-S8 for the adsorption energies, and Table S9 for the parameters of the linear fits). In the light of the theory of scaling relations [19–21,23,24], the scaling relation of \*COH vs \*CH should have a unity slope and a single offset for all facets, in view of their identical valency. Besides, that of \*COH vs \*C should have a slope of 0.75 and a single offset, as well. This is indeed what we observe in Figure 4, where the slopes are 0.75 (\*CH vs \*C), 0.74 (\*COH vs \*C), and 0.99 (\*COH vs \*CH), and all data can be described with a single offset per pair of adsorbates with MAEs of 0.19, 0.26 and 0.16 eV in wide a range of C adsorption energies (> 5 eV).



**Figure 4.** Adsorption-energy scaling relations between \*C (violet), \*CH (blue), and \*COH (cyan) on five surface sites for nine transition metals. Inset: Scaling relation for \*COH vs \*CH. The linear fits and the correlation coefficients (r) are provided in each case. The respective MAEs between the lines and the calculated datapoints are 0.19, 0.26 and 0.16 eV for \*CH vs C, \*COH vs \*C, and \*COH vs \*CH.

#### 3.3. Possible implications for CO<sub>2</sub> electroreduction

For the past decade, adsorption-energy scaling relations have been widely applied for the computational discovery and optimization of (electro)catalysts [9,16,20,21]. While initially it was observed that they simplify screening routines for the design of catalytic materials, it was later noted that they also impose thermodynamic limitations to efficiency of catalysts. This led to the notion of "intrinsic", scaling-based overpotential in computational electrocatalysis [19,45,46]. With the scaling between \*C, \*COH and \*CH in mind, we noted that those three adsorbates have repeatedly been listed as intermediates of the CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) to CH<sub>4</sub> [35,36,41,47–50] .\*COH can be formed from \*CO hydrogenation, and the following two subsequent steps can follow:

$$*COH + H^+ + e^- \rightarrow *C + H_2O \tag{3}$$

$$^{*}C + H^{+} + e^{-} \rightarrow ^{*}CH + H_{2}O \tag{4}$$

The addition of Equations 3 and 4 shows that \*COH and \*CH are separated by two electrons (\* $COH + 2H^+ + 2e^- \rightarrow *CH + H_2O$ ), and Figure 4 shows that they scale with a slope close to 1. The equilibrium potential of the CO<sub>2</sub>RR to CH<sub>4</sub> ( $CO_2 + 8H^+ + 8e^- \rightarrow CH_4 + 2H_2O$ ) is 0.17 V and the free energy of the reaction is -1.36 eV. In this order of ideas, in an ideal CO<sub>2</sub>RR catalyst each proton-electron pair is transferred involving reaction energies of -0.17 eV. Therefore, \*COH and \*CH should ideally be separated by 0.34 eV, and \*C should be exactly in between, at 0.17 eV.

In Figure 5 we show that the calculated separation between \*CH and \*COH is 0.06 eV (note that it is 2.19 eV in Figure 4, because the gas-phase reference is different, see the Methods section and section S5), so that on average there is a mismatch of 0.28 eV (see Figure S3 of the SI for details about the adsorption energies of \*COH, \*C, and \*CH on the considered surfaces and deviations with respect to average values). While this could lead to a scaling-based overpotential of ~0.14 V [19,47], Figure 5 suggests that the location of \*C with respect to \*COH can lead to considerably larger overpotentials. A second version of Figure 5, containing the standard deviations for the individual intermediate energies, can be found in the SI (Figure S4). Indeed, \*COH hydrogenation to produce \*C (Equation 3) is generally an endothermic process and may require significant overpotentials.

This is important to note, as it is usually considered that \*CO hydrogenation to either \*CHO or \*COH is the potential-limiting step of CO<sub>2</sub>RR to CH<sub>4</sub> [16,33–36,41,47–50] Once that step is optimized, probably the next target would be \*COH hydrogenation for materials in which: (i) \*COH is more readily formed than \*CHO, which depends on the metal site, its coordination, and the adsorbate-solvent interactions [35]. ii) \*C is more stable than \*CHOH. Optimizing \*COH hydrogenation might prove challenging, as \*C and \*COH are shown here to bind similarly to transition metal surfaces.

We note that we have focused here on the steps following the formation of \*COH, due to the general interest in higher  $C_1$  reduction products, such as methane. A comprehensive analysis of CO<sub>2</sub> reduction to CO can be found elsewhere [51]. Note in passing that in Figure 5 we averaged all values in two different ways, namely by facet (main panel) and by element (see the inset), so as to condense all of the data and ease their visualization. Combining the results in the main panel and the inset, we conclude that \*COH hydrogenation to \*C will be particularly difficult for (111) surfaces of Cu and noble metals such as Ag, Au, and Pt, with and without step edges. On the other hand, \*COH hydrogenation to \*C seems particularly favorable at (100) terraces, which warrants future analysis: on the one hand, an easy formation of \*C is desirable for efficient CO<sub>2</sub>RR to e.g. CH<sub>4</sub>. On the other hand, the formation of \*C combined with low surface diffusion barriers might lead to coking and, hence, electrode deactivation.



**Figure 5.** Reaction coordinate for the sequential hydrogenation of \*COH to \*C and \*CH on late transition metal surfaces. In the y-axis the free energies are presented with respect to \*COH. The values in black labelled as "ideal" for \*CH and \*COH come from thermodynamic considerations, see the text. The data in colors labelled as "real" or labelled with facets are averages from Tables S3-S8 duly converted into free energies (see section S6). Inset: Free energies with respect to \*COH, averaged per metal.

#### Conclusions

From our simulations it becomes apparent that the adsorption-energy scaling relations of \*CH vs \*C and \*COH vs \*C on transition metal surfaces show an unexpected structural independence. In particular, the offset of such scaling relations is relatively constant across coordination numbers, which so far had only been observed for adsorbates with equal valency, such as \*OH and \*OOH. This leads to three main conclusions:

• Firstly, the adsorbate-surface bonding of \*C, \*COH and \*CH is similar on an electronic scale. Further research into finding an understanding at the orbital level of the \*C, \*COH and \*CH bonding on transition metal surfaces is warranted. This could be achieved by analyzing the density of states, via charge-density plots, using descriptors such as the electron-localization function (ELF) and the (integrated) crystal orbital overlap/Hamiltonian populations (COOP and COHP), and by carrying out an energy-decomposition analysis and/or a principal-component analysis [27,52–55].

• Secondly, the adsorption energy of \*C is not independent of those of \*CH and \*COH, which suggests that that there will be a scaling-relation-based limit to the efficiency of electrocatalysts in reactions that involve those three intermediates. This merits serious consideration when designing new catalysts for reactions pertaining to the carbon cycle.

• A particular case of interest is the CO<sub>2</sub>RR to CH<sub>4</sub>, for which \*COH, \*C and \*CH are likely intermediates. Currently the hydrogenation of \*CO to \*CHO or \*COH is thought to be the potential-limiting step, but once this step is optimized, \*COH hydrogenation is likely the next candidate for optimization, which might be complicated in view of the peculiar structural sensitivity of the adsorption-energy scaling relations shown in this paper.

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#### **Competing financial interests**

The authors declare no competing financial interests.

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